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## Dye-sensitized solar cells with quasi-solid-state cross-linked polymer electrolytes containing aluminum oxide

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#### **1. Introduction**

Dye-sensitized solar cells (DSSCs) have been considered one of the promising alternatives to conventional solar cells, because of their low cost, easy fabrication and relatively high energy conversion efficiency [\[1,2\].](#page-4-0) Based on a liquid electrolyte, a conversion efficiency of 11% for DSSCs has been achieved [\[3\].](#page-4-0) However, the potential problems caused by the liquid electrolytes, such as leakage or evaporation of the solvents, are considered critical problems that limit the long-term operation and practical use of DSSCs. Therefore, considerable efforts have been made to replace the liquid electrolyte with polymer electrolytes [\[4–10\], i](#page-4-0)norganic p-type semiconductors [\[11\]](#page-4-0) and organic hole transport materials [\[12,13\].](#page-4-0) Among these approaches, use of gel polymer electrolytes (GPEs) that exhibit high ionic conductivity at room temperature appears to provide successful results in terms of conversion efficiency. However, such GPEs suffer from poor penetration of electrolyte solution into the porous  $TiO<sub>2</sub>$  electrode and poor interfacial contacts with electrodes. From a practical point of view, these problems can be overcome by cross-linking after the injection of an electrolyte containing cross-linking agent into the cell.

In this work, we synthesized cross-linked GPEs containing aluminum oxide nanoparticles, which allow assembly of quasisolid-state DSSCs by in situ chemical cross-linking after the

#### **ABSTRACT**

Cross-linked gel polymer electrolytes containing aluminum oxide nanoparticles are prepared by in situ chemical cross-linking after injection of the gel precursor into the dye-sensitized solar cell (DSSC). This makes it possible to directly solidify the electrolyte in the cell and maintain good interfacial contacts between the electrolyte and the electrodes without suffering loss of performance in the DSSC. These gel polymer electrolytes exhibit high ionic conductivity and favorable charge transfer reaction at the interfaces of electrodes and electrolyte. The quasi-solid-state DSSC assembled with optimized gel polymer electrolyte exhibited remarkably high conversion efficiency, 6.34% at 100 mW cm−2, and better long-term stability, as compared to the DSSC with liquid electrolyte.

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penetration of the gel electrolyte precursor into the pores of the  $TiO<sub>2</sub>$  electrode. As cross-linking agents, we used poly(ethylene imine) (PEI) with reactive amine groups and poly(ethylene glycol) diglycidyl ether (PEGDE) with reactive epoxy groups. The cross-linking reaction of PEI with PEGDE has been previously reported for applications in lithium batteries [\[14\]. T](#page-4-0)he photovoltaic performance and long-term stability of quasi-solid-state DSSCs assembled with these cross-linked GPEs are investigated and compared with those of liquid electrolyte-based DSSC. A further focus of this study is on the influence of aluminum oxide content on the photovoltaic performance of DSSCs.

#### **2. Experimental**

Gel electrolyte precursor consisting of cross-linking agents, liquid electrolyte and  $Al_2O_3$  particles was prepared as follows. Proper quantities of PEI ( $M_n$  = 423, Aldrich) and PEGDE ( $M_n$  = 526, Aldrich) were dissolved in liquid electrolyte. The liquid electrolyte used in this study is a solution of 0.5 M lithium iodide (LiI),  $0.05$  M  $I_2$  and  $0.05$  M 4-tert-butylpyridine (TBP) dissolved in a mixed solvent of ethylene carbonate(EC)/propylene carbonate(PC) (50:50 by volume). The ratio of PEI to PEGDE in the cross-linking agent was varied, i.e., 1:1, 1:2 and 1:3 by weight, and the cross-linking agents were added at 4.0 wt.% into the liquid electrolyte. Different amounts (0, 5, 10, 15, 20, 25 wt.%) of aluminum oxide (particle size: 2–4 nm, Aldrich) were added to the gel electrolyte precursor, and the mixture was sonicated to ensure homogeneity. Nanocrystalline  $TiO<sub>2</sub>$  paste (Ti-Nanoxide

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<span id="page-1-0"></span>T20/SP, Solaronix) was cast onto a fluorine-doped tin oxide (FTO) glass using a doctor blade and was then sintered at  $450^{\circ}$ C for 30 min. It was sensitized overnight with a cis-diisothiocyanatobis(2,2 -bipyridyl-4,4 -dicarboxylato)ruthenium (II) bis(tetrabutyl ammonium) (Ruthenium 535 bis-TBA, Solaronix) dye solution. Counter electrode was prepared by spin-coating 0.01 M  $H_2PtCl_6$  in isopropanol onto the FTO glass and sintering at 450 ◦C for 30 min. The cell was fabricated by sealing the  $TiO<sub>2</sub>$  electrode and Pt counter electrode together using a hot melting film. Then, the gel electrolyte precursor was injected into the cell through a hole in the Pt electrode. The holes were then completely sealed by a sealant. Finally, the assembled cell was placed in a  $50^{\circ}$ C oven for 2 h to induce the chemical cross-linking reaction. Photovoltaic performance of DSSCs was evaluated using a xenon light source (100 mW cm−2) with an AM 1.5 filter in a solar simulator at ambient temperature. The light intensity was calibrated with a NREL-calibrated Si solar cell (PV Measurements Inc.). A black mask of  $0.25 \text{ cm}^2$ aperture was placed over the cells during irradiation and an antireflection glass was placed on the front glass cover of the cells. For long-term stability test, the cells were stored in a desiccator at room temperature and applied to electrochemical measurements every 24 h. A two-electrode electrochemical cell consisting of crosslinked GPE sandwiched with two identical Pt electrodes was used to measure the diffusion coefficient of the triiodide ion from the diffusion-limited current measurements [\[15,16\].](#page-4-0) The diffusionlimited current density was determined by cyclic voltammetry using a scan rate of 5 mV s<sup>-1</sup>. In order to investigate the impedance behavior of DSSCs, AC impedance measurements were performed using an impedance analyzer (CH Instruments) over the frequency range of 10 mHz to 100 kHz at open circuit under 1 Sun illumination. The applied bias voltage and ac amplitude were set at the open-circuit voltage and 10 mV, respectively. Impedance spectra were analyzed by an equivalent circuit model for interpreting the characteristics of the DSSCs [\[17–19\].](#page-4-0)

#### **3. Results and discussion**

The cross-linking reaction was performed in an electrolyte solution containing iodine without harmful initiators at 50 ◦C, which prevents the formation of residual impurities in the GPE. After cross-linking for 2 h, the electrolyte becomes non-fluidic, due to the formation of three-dimensional polymer networks, as shown in the schematic diagram of the quasi-solid-state DSSC (Fig. 1).



**Fig. 2.** Photocurrent–voltage curves for DSSCs assembled with gel electrolyte precursor before and after cross-linking reaction, as a function of ratio of cross-linking agent (PEI: PEGDE): (a) before cross-linking reaction, (b) after cross-linking reaction.

The photovoltaic performance of DSSCs assembled with gel electrolyte precursor without  $Al_2O_3$  nanoparticles was evaluated. Fig. 2 represents the photocurrent density–voltage curves of the DSSCs with gel electrolyte precursor before and after cross-linking reaction, as a function of ratio of PEGDE to PEI in the cross-linking agents. For comparison, the photocurrent density–voltage curve of



Fig. 1. Schematic diagram of DSSC before and after in situ chemical cross-linking reaction of gel electrolyte precursor containing Al<sub>2</sub>O<sub>3</sub>.

<span id="page-2-0"></span>

**Fig. 3.** Ionic conductivities and diffusion coefficients of  $I_3$ <sup>-</sup> in the cross-linked gel polymer electrolyte (PEI: PEGDE = 1:3 by weight), as a function of  $Al_2O_3$  content.

the DSSC prepared with liquid electrolyte is also shown in [Fig. 2.](#page-1-0) In [Fig. 2\(a](#page-1-0)), the open circuit voltages ( $V_{\text{oc}}$ ) of the DSSCs with gel electrolyte precursor are higher than that of the liquid electrolytebased DSSC. The increase of  $V_{\text{oc}}$  may result from the basicity of PEI, because the flat band potential of  $TiO<sub>2</sub>$  shifts negatively with the increase of the basicity of the electrolyte, which results in an increase of  $V_{\text{oc}}$  [\[20\]. T](#page-4-0)he addition of gel electrolyte precursor into the liquid electrolyte also causes a slight decrease of short circuit current density  $(J_{\rm sc})$  of DSSCs, which is due to the increase in viscosity of electrolyte. After cross-linking reaction with gel electrolyte precursor, the  $V_{\text{oc}}$  values of DSSCs were hardly changed, as shown in [Fig. 2\(b](#page-1-0)). On the other hand, the values of  $J_{\rm sc}$  of DSSCs decreased after cross-linking reaction. This is due to the fact that the crosslinking causes an increase in the resistance for ion migration in both the electrolyte and the electrode. It should be noted that  $J_{\rm sc}$  slightly increases with increasing relative ratio of PEGDE to PEI. PEGDE itself has a high dielectric constant ( $\varepsilon$  = 13.7) and a low glass transition temperature, which provides both generation of higher charge carriers ( $I^{-}/I_{3}^{-}$ ) and fast ion migration in the DSSCs with high PEGDE content. As a result, the cell assembled with PEI: PEGDE = 1:3 exhibited the highest conversion efficiency (4.82%). This composition was thus chosen for assembling the quasi-solid-state DSSCs for further studies.

Addition of inorganic nanoparticles, such as  $SiO<sub>2</sub>$  and TiO<sub>2</sub> into the electrolyte is an effective way to improve the photovoltaic performance of DSSCs assembled with gel electrolytes [\[21–24\]. I](#page-4-0)n this work, we introduced nanosized  $Al_2O_3$  particles into the crosslinked GPEs. Before applying the cross-linked GPEs into the cell, we measured ionic conductivities and diffusion coefficients of  $I_3^-$ . The diffusion coefficient of  $I_3^-$  in the cross-linked GPE was estimated by measuring the diffusion-limited current density in the steadystate current–voltage curve [\[15,16\].](#page-4-0) Fig. 3 shows the variation in ionic conductivities and diffusion coefficients of  $I_3^-$  in the GPEs as a function of  $Al_2O_3$  content. Both ionic conductivity and diffusion coefficients increase with increasing content of  $Al_2O_3$  nanoparticles, reaching a maximum at 20 wt.%, followed by a decrease. In a previous study, Huo et al. reported that the diffusion coefficient of  $I_3^-$  reached a maximum at 10 wt.% of TiO<sub>2</sub> in the poly(vinylidene fluoride-co-hexafluoropropylene)-based nanocomposite GPE [\[24\].](#page-4-0) An increase in ionic conductivity and diffusion coefficient with the addition of  $Al_2O_3$  is related to the generation of free volume at the nanoparticle interface [\[23\]. A](#page-4-0)s also previously reported [\[24,25\],](#page-4-0) charge can transport effectively in the channels of the nanocomposite gel network, and a special mechanism (a Grotthuss-like exchange mechanism) may contribute to the effective charge transport. A decrease in ionic conductivity and diffusion coefficient beyond 20 wt.%  $Al_2O_3$  may be attributed to the blocking effect on the transporting of charge carriers by the aggregation of  $Al_2O_3$ nanoparticles. The morphologies of cross-linked nanocomposite gel polymer electrolytes were analyzed by SEM images shown in Fig. 4, which were obtained after removing the electrolyte solution. It is seen that the porosity of the nanocomposite gel polymer electrolyte increases with addition of  $Al_2O_3$ . This result indicates that the  $Al_2O_3$  nanoparticle generates a lot of pores, which are filled



Fig. 4. Scanning electron micrographs of the surface of the cross-linked gel polymer electrolytes containing different contents of Al<sub>2</sub>O<sub>3</sub>: (a) 0 wt.% Al<sub>2</sub>O<sub>3</sub>, (b) 10 wt.% Al<sub>2</sub>O<sub>3</sub>, (c) 20 wt.%  $Al_2O_3$ , (d) 25 wt.%  $Al_2O_3$ .

<span id="page-3-0"></span>

Fig. 5. Effect of Al<sub>2</sub>O<sub>3</sub> content on the photovoltaic performances of DSSCs assembled with cross-linked gel polymer electrolytes: (a) photocurrent–voltage curves for DSSCs, (b)  $J_{\rm sc}$  and  $\eta$  vs. Al<sub>2</sub>O<sub>3</sub> content.

with liquid electrolyte solution. However, the porosity is observed to decrease at 25 wt.%  $Al_2O_3$ . From these results, it is believed that the addition of 20 wt.%  $Al_2O_3$  provides the most desirable environment for ionic transport in the cross-linked nanocomposite GPE under study. At the optimum 20 wt.%  $Al_2O_3$ , the ionic conductivity and the diffusion coefficient of  $I_3^-$  are  $6.0 \times 10^{-3}$  S cm<sup>-1</sup> and  $3.8 \times 10^{-6}$  cm<sup>2</sup> s<sup>-1</sup>, respectively.

The photovoltaic performance of the DSSCs assembled with cross-linked GPE containing  $Al_2O_3$  particles was evaluated, and the results are shown in Fig. 5. The  $J_{\rm sc}$  and conversion efficiency  $(\eta)$ of the DSSC increased with the addition of  $Al_2O_3$  up to 20 wt.% and decreased with the further addition. An increase in  $J_{\rm sc}$  with Al $_2$ O $_3$  content is related to improved diffusion of I $_3$ –, as explained in [Fig. 3.](#page-2-0) In Fig. 5(a), the increase in  $V_{\text{oc}}$  is hardly seen with addition of  $Al_2O_3$ , indicating that the  $Al_2O_3$  has very minor influence on  $V_{\text{oc}}$ . The optimum  $Al_2O_3$  content to achieve the best conversion efficiency is about 20 wt.%, as shown in Fig. 5(b). Notably, the conversion efficiency (6.34%) of the DSSC assembled with cross-linked GPE containing 20 wt.%  $Al_2O_3$  is higher than that of the corresponding liquid electrolyte-based DSSC (5.97%).

In order to investigate the effect of  $Al_2O_3$  on the impedance behavior of the DSSCs, the ac impedance of the cells was measured. The resulting ac impedance spectra are shown in Fig. 6. For comparison purpose, the ac impedance spectrum of the DSSC prepared with liquid electrolyte is also shown. As shown, the spectra exhibit three overlapping semicircles, which can be assigned to the electrochemical reaction at the Pt counter electrode  $(R<sub>ct1</sub>)$ , the charge



**Fig. 6.** AC impedance spectra of DSSCs assembled with liquid electrolyte and crosslinked gel polymer electrolyte containing different contents of  $Al_2O_3$ .



**Fig. 7.** Variation of short-circuit current density and conversion efficiency as a function of time for the DSSCs assembled with liquid electrolyte and cross-linked gel polymer electrolytes containing 20 wt.%  $Al_2O_3$ : (a) short-circuit current density vs. time, (b) conversion efficiency vs. time.

<span id="page-4-0"></span>transfer reaction at the TiO<sub>2</sub> electrode ( $R_{ct2}$ ) and the Warburg diffusion process of  $I^-/I_3^-$  ( $R_{\rm diff}$ ), respectively. The electrolyte resistance  $(R_e)$  can be estimated from the intercept on the real axis in the highfrequency range. After cross-linking reaction without  $Al_2O_3$ , all of the resistances increased as compared to liquid electrolyte-based DSSC. As mentioned earlier, this result arises from the increase in the resistance for ion migration in both the electrolyte and the electrode after cross-linking reaction. In the DSSCs assembled with the cross-linked GPE containing  $Al_2O_3$ , the electrolyte resistance is decreased with addition of  $Al_2O_3$  and shows the lowest value at 20 wt.%  $Al_2O_3$ , which is consistent with the ionic conductivity results in [Fig. 3.](#page-2-0) Total interfacial resistance estimated from  $R_{ct1}$  and  $R_{ct2}$  also reached a minimum at 20 wt.%  $Al_2O_3$ . The presence of nano-sized  $Al_2O_3$  in the GPE increases the interfacial area and thus enhances the charge transport at the interface between the electrolyte and the electrode. These results indicate that the addition of proper amount of  $Al_2O_3$  leads to an improvement in the charge transfer reaction at both the  $TiO<sub>2</sub>$  electrode and the Pt counter electrode, as well as enhancement of transport of  $I^-/I_3^-$  ions in the electrolyte, which results in better cell performance.

The long-term stability of the quasi-solid-state DSSC assembled with cross-linked GPE containing 20 wt.%  $Al_2O_3$  was evaluated. [Fig. 7](#page-3-0) represents the variation of  $J_{\rm sc}$  and conversion efficiency of the quasi-solid-state DSSC as a function of time. For comparison, the J<sub>sc</sub> and conversion efficiency of the DSSC with liquid electrolyte are also shown. The conversion efficiency in the liquid electrolytebased DSSC decays continuously with time during the period of investigation. After 10 days, the cell lost 48% of its initial conversion efficiency. Gradual decrease of  $J_{\rm sc}$  and conversion efficiency in the DSSC assembled with liquid electrolyte is related to the evaporation of liquid electrolyte, which is caused by the imperfect sealing of the DSSC. On the other hand, the conversion efficiency of the quasisolid-state DSSC assembled with cross-linked GPE remains almost constant. In the case of the DSSC assembled with chemically crosslinked GPE, the organic solvent containing the I $^-/$ I $_3^-$  redox couple is well encapsulated in the cell, and the in situ chemical cross-linking can also promote good interfacial contact between a dye-adsorbed  $TiO<sub>2</sub>$  electrode and a platinum counter electrode, which give more stable performance than that of the DSSC assembled with liquid electrolyte.

#### **4. Conclusions**

Quasi-solid-state DSSCs were fabricated by in situ chemical cross-linking of gel electrolyte precursor containing  $Al_2O_3$ nanoparticles, and their photoelectrochemical performance was evaluated. The addition of an appropriate amount of  $Al_2O_3$ nanoparticles into the cross-linked gel polymer electrolyte improved the conversion efficiency of DSSCs. The quasi-solid-

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